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Room Temperature Uniaxial Magnetic Anisotropy Induced By Fe-Islands in the InSe Semiconductor Van Der Waals Crystal

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Magnetic anisotropy is at the heart of spintronics: it gives rise to an energy barrier between two opposite spin directions and hence to the possibility to store and process information. Magnetic anisotropy has been demonstrated in several material systems, ranging from magnetic ion nanoclusters to single-molecule magnets to single atoms. However, the integration of magnetic and semiconducting properties within the same material system is generally difficult to achieve. This integration is important for several applications and new device concepts in spintronics, including electrical control of the magnetization and generation of spin-currents, spin-filtering and spin-amplification in logic devices. To date, the realization of magnetic semiconductors has proven to be challenging and it has led to material systems with interesting magnetic and electronic properties.

In this work, we demonstrate magnetic and semiconducting properties in the van der Waals (vdW)-layered crystal InSe, a material system that has emerged as a promising candidate for electronics and photonics due to its high electron mobility, chemical stability, and high photoresponsivity. In a vdW crystal, the atoms in each layer are bound by strong covalent bonds, whereas the planes...
are held together by weak vdW interactions. The extended family of vdW crystals includes graphene, hexagonal boron nitride, transition metal dichalcogenides, and many others. Although a variety of semiconductor crystals and stacks has been demonstrated, the available structures are nonmagnetic, weakly magnetic, or magnetic only at low temperature. Here, we show that InSe, which is nonmagnetic in its pristine form, becomes magnetic following the incorporation of Fe-atoms during the growth of InSe by the Bridgman method. We show that the Fe-atoms self-assemble into islands embedded within the InSe host-crystal. The islands are crystalline and have a triangular shape in the plane of the vdW layers. Our material tends to retain the electronic, optical, and vibrational properties of pristine InSe. However, the Fe-islands imprint the InSe crystal with a large uniaxial magnetic anisotropy at room temperature with the magnetization preferentially oriented in the direction perpendicular to the plane of the vdW layers. Thus, room temperature magnetism and semiconducting properties are achieved within the same material system, offering opportunities for further research developments and exploitation.

Figure 1. Fe-islands in InSe van der Waals crystals. a) Schematic representation of the crystal structure of InSe. b) SEM-EDX maps of the surface of an InSe crystal grown by the Bridgman method with a nominal Fe-content of 10%. c) Cross-sectional TEM image and EDX map of an Fe-island in InSe. The island is elongated along the c-axis.

Irrespective of the Fe-content, we find that for all our InSe bulk crystals the energy peak position of the room temperature ($T = 292$ K) photoluminescence (PL) emission is centered at an energy $h\nu = 1.25$ eV (Figure 3a) and the Raman peaks are at 115.7, 179.2, 201.2, 212.4, and 228.0 cm$^{-1}$, as observed for pristine bulk InSe (Figure S2, Supporting Information). With increasing Fe-content, the intensity of the optical signals tends to decrease and spatial maps of the PL intensity reveal an increasing nonhomogeneity over length scales of a few micrometers (Figure 3b). Correspondingly, the room temperature conductivity in the layer plane decreases due to a reduction of the electron mobility from $\mu = 10^3$ cm$^2$ V$^{-1}$ s$^{-1}$ in pristine InSe to $\mu \leq 10^2$ cm$^2$ V$^{-1}$ s$^{-1}$ in the InSe crystals containing Fe. Thus, despite the incorporation of Fe-islands in the $ab$-plane (Figure 1b). In the thicker films (>1 µm), these islands are elongated along the c-axis (Figure 1c). These nanostructures contain a high content of Fe (>95%) and have a triangular shape in the $ab$-plane, as shown in the high-resolution EDX image of Figure 2a. In this figure, the two equal sides of the triangular Fe-islands have length of 0.8 µm. Furthermore, the Fe-atoms are arranged into a body centered cubic (bcc) lattice with lattice constant $a = 2.87$ Å (Figure 2b), as for bulk $\gamma$Fe. Thus, Fe-islands with a bcc lattice (Figure 2b) coexist with the rhombohedral crystal structure of $\gamma$InSe (Figure 2c). The low solubility of Fe in InSe and the large Fe-content create supersaturation conditions during the Bridgman growth of $\gamma$InSe, thus leading to the formation of two distinct crystals within the same material system (Experimental Section).

Magnetic force microscopy (MFM) measurements were conducted in the thermally demagnetized state of the crystals using a two-pass method and a CoCr-tip magnetized along the tip axis. The first pass was conducted in a tapping mode to reveal the surface topography (e.g., atomic force microscopy, AFM); this was then followed by the second pass at an increased scan height $h$ (25–50 nm) to probe long-range magnetic interactions. During the MFM imaging, the scan height $h$ is maintained constant and the changes in the phase of the oscillating probe are recorded.13 These changes originate from the long-range magnetic interactions between the probe and the sample.

Figure 2d,e shows the AFM and MFM images for a thin InSe film (~100 nm) at the locations of the Fe-islands #1 and #2 highlighted in the EDX image of Figure 2a. Dark and bright contrasts in the MFM images (Figure 2e) correspond to magnetic repulsion and attraction, respectively, indicating the presence of perpendicular stray fields emanating from magnetic domains. Both Fe-islands have a similar domain structure suggesting a close similarity in their magnetic properties (Figure 2e, top). The enlarged AFM and MFM images in the bottom of Figure 2d,e show with greater detail the shape, orientation, and height of an individual Fe-island and its threefold multidomain state. By comparing the morphology and the domain structure of this island, we conclude that the domain structure is 3D. The largest domain on the right-hand side morphs around the upper-right-hand edge. The two other domains are similar in size, completing the domain closure in an anticlockwise fashion (see schematic representation in the left bottom corner of Figure 2e). These findings demonstrate that ferromagnetic Fe-islands with a bcc lattice are embedded within the rhombohedral $\gamma$InSe crystal.
InSe, the crystal preserves many of the functional properties of pristine InSe. Furthermore, the crystals can be exfoliated into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL emission peak undergoes a strong blueshift into thin layers and the PL 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Figure 3. Semiconducting properties of InSe containing Fe-islands. a) Energy shift of the room temperature PL peak versus the layer thickness $L$ for InSe layers with nominal Fe-content of 0% (black), 1% (red), and 10% (blue). The solid line is the energy shift calculated using an effective mass quantum well model. Inset: PL spectra for bulk InSe with nominal Fe-content of 0%, 1%, and 10% ($\lambda = 633$ nm; $P = 0.1$ mW). b) Maps of the PL intensity for bulk InSe with nominal Fe-content of 0%, 1%, and 10% ($\lambda = 633$ nm; $P = 0.1$ mW).

We model the measured angular dependence of the resonance linewidth $\Delta B$ (Figure 4d) as

$$\Delta B = \Delta B_0 + c\alpha/\gamma$$  \hspace{1cm} (2)

where $\Delta B_0$ and $c\alpha/\gamma$ are the inhomogeneous and homogeneous broadening, respectively, and $\alpha$ is the dimensionless Gilbert damping parameter.$^{[19]}$

The inhomogeneous linewidth broadening $\Delta B_0$ is attributed to the nonhomogeneous internal magnetic fields arising from the random distribution of the Fe-islands (Figure 1). The Gilbert damping parameter accounts for the losses of spin angular momentum during the precession of the magnetization around an effective magnetic field $B_{eff}$ that includes the external, internal, and microwave field. When $B_{eff}$ is parallel to the $c$-axis, the external magnetic field and the magnetization direction are parallel and $\Delta B$ has a minimum; in contrast, in the out-of-plane rotation of $B_{eff}$, $\Delta B$ first increases by about 10% for $\theta_g = 45^\circ$ and then increases steeply for $\theta_g$ approaching a value of $\theta_g = 90^\circ$ (Figure 4d). The angular dependence of $\Delta B$ suggests a magnetic dragging effect due to the noncollinearity of the external magnetic field and the magnetization direction.$^{[19]}$

The contribution to the linewidth broadening $\Delta B$ of the Gilbert parameter (Equation (2)) implies angular momentum losses of the magnetization precession in the magnetic Fe-islands into the nonmagnetic InSe matrix. This spin-pump mechanism$^{[4]}$ occurs at room temperature, offering prospects for the generation of a charge current in InSe via the inverse spin Hall effect.$^{[20]}$

The uniaxial magnetic anisotropy is observed at room temperature and it depends only weakly on temperature, as assessed by SQUID at low ($T = 5$ K) and room temperature ($T = 300$ K) (Figure S3, Supporting Information). Due to the coexistence of mixed phase states within a system that consists of a diamagnetic InSe crystal and ferromagnetic Fe-islands (Figure 2), the temperature dependence of the magnetic susceptibility tends to be weak and different from that expected for ferromagnetic Fe (Figure S3, Supporting Information). Furthermore, the ESR spectra reveal a complex behavior. The weakening of the main ESR lines, $g_1$ and $g_2$, with decreasing temperature indicates the emergence of an anisotropic antiferromagnetic (AF) order at a Néel temperature $T_N = 260$ K (Figure 5b). An AF order with a Néel temperature $T_N$ up to 100 K was reported for $\gamma$Fe thin films with face-centered cubic (fcc) crystal symmetry obtained by epitaxial growth$^{[21]}$ or precipitation,$^{[22]}$ an AF order can emerge when bcc $\gamma$Fe undergoes a crystal phase transition to fcc below a critical layer thickness$^{[23]}$ due to stronger magnetic interactions arising from a smaller lattice constant$^{[24]}$ and/or surface effects$^{[25]}$; AF and FM orders can also coexist within an Fe-cluster due to its composite crystal structure.$^{[26]}$ The coexistence of different magnetic phases
in our system may arise from nonequivalent Fe-atoms in the islands and strain effects at the interface with the diamagnetic InSe, which requires further investigation.

To conclude, recent advances in the science and technology of vdW crystals have demonstrated the potential of this class of materials for novel functional devices. Among these crystals, InSe has emerged as a semiconducting system with unique electronic and optical properties, including high electron mobility\(^{[11]}\) and strong photosensitivity.\(^{[12]}\) Here, we have shown that the formation of crystalline Fe-islands in InSe induces a uniaxial internal magnetic field (≈1 T) perpendicular to the InSe layers. Thus, this hybrid system, which consists of Fe-inclusions and a van der Waals crystal, enables the coexistence of magnetic and semiconducting properties within the same structure. Our findings will stimulate further research on magnetism in novel semiconductor materials beyond conventional Si\(^{[27]}\) and GaAs.\(^{[28]}\) Since vdW crystals are compatible with other vdW crystals, magnetic metals, and dielectrics, we envisage further developments and a new class of devices that exploit the magnetic properties of hybrid magnetic-semiconducting materials. In particular, losses of spin angular momentum during the precession of the magnetization in the ferromagnetic Fe-islands into the nonmagnetic InSe offer prospects for the generation of a charge current in InSe via the inverse spin Hall effect and its control by the magnetic anisotropy of the crystal. Further developments also include the homogenous incorporation of substitutional Fe-atoms in InSe, which has recently been proposed as a route to create a homogeneous ferromagnetic semiconductor.\(^{[29]}\)

**Experimental Section**

**Materials and Structural Studies:** The crystallization of the Fe-islands occurred along with the formation of the layered InSe vdW crystal during the Bridgman growth. The InSe:Fe melt was contained inside a quartz...
ampoule and was cooled down slowly using a moving crucible inside the Bridgman furnace.

The crystal structure of all crystals was probed by X-ray diffraction using a DRON-3 X-ray diffractometer that used monochromatic Cu Kα radiation of wavelength $\lambda = 1.5418$ Å (Section S1, Supporting Information). Transmission electron microscopy (TEM) experiments were conducted on thin sections of the crystals prepared by focused ion beam scanning electron microscope, FIB-SEM (FEI Quanta 3D). The crystals were studied using a JEOL 2100F microscope operating at 200 kV, equipped with a Gatan Orius camera and Oxford Instruments X-Max 80 EDX detector. SEM EDX studies were performed on the crystals using an FEI Quanta 650 operating at 20 kV, equipped with an Oxford Instruments X-Max 150 detector. The flakes were prepared from the as-grown crystals by mechanical exfoliation using adhesive tape and then transferred onto a Si/SiO$_2$ substrate. Images of the flakes' topography were acquired using an Asylum Research MFP-3D AFM operated in tapping mode under ambient conditions.

**Magnetic and Optical Studies:** MFM measurements were performed on a Dimension Icon (Bruker) scanning probe microscope (SPM). The MFM probe (Nanosensor PPP MFMR) had a typical spring constant $k = 2–3$ N m$^{-1}$ and curvature radius $r < 30$ nm.$^{[90]}$ The MFM imaging was carried out using a two-pass technique. During the first pass, the SPM was operated in the atomic force microscopy mode to determine the topography. During the second pass, the topography line (obtained during the first pass) was retraced while oscillating the probe at a frequency $f = 69.56$ kHz (free-space amplitude $A_f \approx 200$ nm), maintaining a set distance of $h = 25$ nm between the probe and sample, and recording the cantilever phase change due to probe–sample magnetic interactions. The scan across the sample was conducted at a rate of 0.7 Hz.

ESR measurements were recorded on commercial Bruker EMX and E580 spectrometers operated at Q-band (34 GHz). The microwave field was perpendicular to the external magnetic field B (perpendicular mode). Typical experimental conditions were as follows: modulation amplitude of 0.5 mT, modulation frequency of 50 kHz, and conversion time of 20 ms. The experimental setup for the micro-PL and Raman studies comprised a He-Ne laser ($\lambda = 633$ nm) or a frequency doubled Nd:YVO$_4$ laser ($\lambda = 532$ nm), an x-y-z motorized stage and an optical confocal microscope system equipped with a 0.5 meter long monochromator with a 150 and 1200 g mm$^{-1}$ gratings. The laser beam was focused to a diameter $d = 1$ μm using a 100x objective. Optical experiments were performed at low excitation power ($P < 0.1$ mW) to avoid excessive heating. The signal was detected by a Si charge-coupled device camera. Magnetometry was performed on a commercial Quantum Design MPMS 3 SQUID.

**Figure 5.** Magnetic phases induced by Fe-islands in InSe. a) Temperature-dependent ESR spectra for an InSe crystal with nominal Fe = 10% (Q-band and B parallel to the c-axis). Inset: Orientation of B relative to the crystallographic c-axis of InSe. The ESR spectra are shifted along the vertical axis for clarity. Temperature values from top to bottom are $T = 292, 250, 200, 150, 120, 100, 80, 60, 50, 40, 30, 20, 10,$ and $5$ K. b–d) ESR intensity, resonance field $B_{res}$ and linewidth $\Delta B$ of resonance $g_2$ versus T. The inset in part (b) sketches the low $T$ antiferromagnetic order. The inset in part (c) shows the correlation between the resonance linewidth and resonance field.
Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

F.M. and A.P. designed the experiments and wrote the paper; all authors took part in the discussion and analysis of the data. In particular, Z.R.K. and Z.D.K. grew the bulk crystals and prepared the exfoliated layers; M.W.F. conducted the TEM studies; C.P. conducted the EDX studies; F.M., M.A.B., and A.J.F. performed the ESR experiments; M.K. and J.V.S. conducted the SQUID experiments; R.P. and O.K. conducted the MFM studies; M.A.B., Z.R.K., O.M., and A.P. conducted the optical and electrical measurements. This work was supported by the EU Graphene Flagship Project; the Engineering and Physical Sciences Research Council (grant number EP/M012700/1); the Royal Society; the Leverhulme Trust; the EPSRC National EPR Facility at The University of Manchester [grant number NS/A000014/1] the European Metrology Programme for Innovation and Research (grant number 1551806); the NanoMag; the Center for Integrated Quantum Science and Technology. The authors acknowledge useful discussions with A. Brookfield, M. Perfetti, C.R. Staddon, and N. Balakrishnan.

Conflict of Interest

The authors declare no conflict of interest.

Keywords

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